

Quantum size effects in Pb/Si(1 1 1) thin films from density functional calculations

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ABSTRACT

The Pb/Si(1 1 1) thin films were simulated within the density functional theory (DFT). The well-known Perdew–Burke–Ernzerhof (PBE) version of the generalized gradient approximation (GGA) and its recent nonempirical successor Wu–Cohen (WC) issue were used to estimate the exchange–correlation functional. Lattice parameters were calculated for bulk of the Pb and Si compounds to obtain more reliable lattice mismatch at the interface to be consistent with our used full-potential method of calculations. The WC–GGA result predicts the lattice constants of the Pb and Si compounds better than the GGA when compared with experiment. We have found that the spin–orbit coupling (SOC) does not significantly influence the results. Our finding is in agreement with the recent observation of the Rashba-type spin–orbit splitting of quantum well states in ultrathin Pb/Si(1 1 1) films. Our result shows, in agreement with experiment, that the top site (T1) is the most stable phase. A combination of tight σ and feeble π bonds has been found at the interface, which results in a covalent Pb–Si bond. Our calculated electric field gradient (EFG) predicts quantum size effects (QSE) with respect to the number of deposited Pb layers on the Si substrate. The QSE prediction shows that the EFG dramatically drops on going from first to second layer. The EFG calculation shows that this system is not an ideal paradigm to freestanding films.

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1. Introduction

The Pb/Si(1 1 1) thin films can be considered as an ideal interface, since it is an unreactive metal–semiconductor system which forms perfectly an abrupt interface without intermixing [1]. Intensive experimental [2,3] and theoretical [4,5] works performed on this system show that it is a subject of intense technological and scientific interest with a wide range of applications. Several structures were proposed experimentally for this system [1,6]. The structures vary with their circumstances. The variations form a complicated phase diagram. The stability of the proposed structural models were studied theoretically employing pseudopotential calculations [4]. The complex phase diagram shows that there are four structures with a simple (1×1) -unit cell. The experimentally observed [7] simple unit cell provides a suitable case to perform accurate atomic relaxations within an all-electron calculation reliably. Our full-potential surface calculations were then devoted to study these (1×1) structures with the concentration on their most stable phase. Quantum size effects (QSEs) in nanostructures, e.g., a two-dimensional (2D) ultrathin metal films, appear as dramatic oscillations in many physical properties upon variation of film thickness [8–11]. The motion of electrons in the film plane of such a 2D-system is essentially free, whereas the electrons are con-

finied in the normal direction to the film surface, which leads to the quantized electronic states, i.e., quantum well states (QWS) [12]. The oscillation period was found to be 7-monolayer-height-islands of lead grown on-top of a silicon substrate for the electron intensity curve as a function of the normal component of the electron momentum transfer [8]. Vázquez de Parga and coworkers observed the QSEs for a number of surface properties, i.e., the surface roughening temperature, work function, chemical reactivity, or the surface diffusion barrier, in the Pb/Si(1 1 1) and Pb/Cu(1 1 1) thin films by Scanning Tunneling Microscopy/Spectroscopy (STM/STS) [9]. The step height of layer N is related to the electron spillage length into the vacuum which depends on the DOS at the Fermi level [9]. Quantum size effects (QSEs) were found for the total energies and energy differences in freestanding Pb(1 1 1) thin films by using pseudopotential method as embodied in the CASTEP code [13]. Influence of the Cu(1 1 1) substrate on the geometry structure of the Pb(1 1 1) layers was also indirectly taken into account by solely reducing in-plane lattice constant to impose 3.3% lateral compression on all the Pb(1 1 1) layers to mimic the role of Cu(1 1 1) substrate [13]. The effects of Cu(1 1 1) substrate were found to be more substantial, as the step heights were calculated to be in more agreement with experiment within the later in-plane strain [13]. Wei and Chou theoretically, employing ultrasoft pseudopotential calculations, studied the quantum size effects (QSEs) in the clean surface of Pb(1 1 1) [14]. Recently Kirchmann et al. experimentally observed quantum size effects (QSEs) in the Pb/Si(1 1 1) [15]. They

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[15] found their experimental results for the actual Pb/Si(111) slab to be in agreement with the pseudopotential results [14] for the hypothetical freestanding Pb(111) slab. Dil et al. [16] comparing their Pb/C full-potential calculations with the Pb/Si(111) experimental results of the others [17] reported that unlike for Pb on graphite, the Pb overlayer lattice structure is influenced by that of the Si(111). Here we then intend to, going beyond the hypothetical freestanding Pb(111) approximation, quantitatively assess the effects of electronic and crystalline structures of the underlying Si substrate on the deposited Pb films. For this reason we inspect the electronic structures at the interface of the Pb/Si(111) slab. From our electronic structure calculations, we show that there is a mixed state composed of weak π and strong σ bonds between Pb and Si at the interface which results in a strong Pb–Si covalent bond. We aim to investigate whether the substrate can influence the results. In order to accomplish the investigation, we have calculated the work function, energy differences and surface formation energy as a function of number of Pb layers for the (1×1) -Pb/Si(111). Our result shows that the effect of Si substrate can be of significant importance for some physical quantities depending on their sensitivity to the valence electron charge density. We have observed the quantum size effects (QSEs) for the (1×1) -Pb/Si(111) thin films. The goal of this work more transparently is achieved by presenting a physical interpretation for our QSE calculations for this system in the electric field gradient (EFG) as an extremely sensitive quantity to the valence electron charge density distribution. We would also examine the effect of spin–orbit coupling (SOC) on our thin films. The result, in agreement with experiment [18], shows that the SOC has a minor effect for the Pb/Si(111) thin films. Low temperature $\sqrt{3} \times \sqrt{3}$, 3×3 and $\sqrt{7} \times \sqrt{3}$ phases of Pb/Si(111) were theoretically studied employing pseudopotential method [19]. It was shown that there could be a discrepancy in predicting the ground state of Pb/Si(111) system between local density approximation (LDA) and generalized gradient approximations (GGA) [19]. Another objective of our density functional theory [20,21] (DFT) calculations is to go beyond the LDA and GGA by using the nonempirical GGA recently proposed by Wu–Cohen (WC) for the exchange–correlation functional [22], which is expected to improve metal surface formation energy and lattice constant calculations [22,23] compared to the Perdew–Burke–Ernzerhof GGA (PBE-GGA) [24].

2. Computational details

State-of-the-art calculations were performed within the density functional theory [20,21] (DFT) as implemented in the WIEN2k code [25]. The augmented plane waves plus local orbital (APW+lo) method [26,27] has been used for solving the Kohn–Sham equations [21] employing the latest version of the generalized gradient approximation (GGA), i.e., Wu–Cohen (WC), for the exchange–correlation functional [22]. We set the Muffin-tin radii to $R_{MT} = 2.2$ bohr for the Si and to $R_{MT} = 2.5$ bohr for the Pb atoms. The expansion of the wave functions and charge densities were cut off by the $R_{MT}K_{max} = 7.5$ and $G_{max} = 14$ parameters, respectively. The full relaxations were performed with the criterion of 1 mRy/bohr on the exerted forces. The relativistic effects were taken into account by including the spin–orbit coupling (SOC) in a second variational procedure. A set of $22 \times 22 \times 1$ special k points has been used for integrations over the Brillouin zone of the 1×1 surface cell.

3. Bulk structural properties

Before going through the interface and surface calculations in detail, it seems advisable to first shortly report on the bulk structural properties. In this case, the accuracy of our calculations could

be first assessed, and second more accurate lattice parameters would be also obtained to simulate the substrate with an actual and reliable lattice mismatch at the interface. Furthermore one aims to get more insight into the various exchange–correlation functionals, specially the recent proposed WC-GGA [22] potential. The lattice parameters, bulk moduli, and the pressure derivative of the bulk moduli were calculated for the bulks of the Si and Pb compounds. The calculations were performed within the LDA [28], PBE-GGA [24] and WC-GGA [22] functionals by fitting the total energy as a function of volume with the Birch equation of state [29]. The current APW+lo and previous pseudopotential plane wave (PPW) results [30–32] along with the experimental data [33,34] are presented in Table 1. Our APW+lo result, compared to the experiment [33,34], shows that, for both the Si and Pb cases, the PBE-GGA [24] overestimates the lattice constants and bulk moduli, whereas the LDA [28] underestimates them. This is consistent with the general features of the PBE-GGA [24] and LDA [28] functionals for solid states. In particular, as shown in Table 1, our values are in complete accord with the previous PPW results [30–32]. The WC-GGA [22] predicts the lattice constant of the Pb compound better than the LDA [28] and GGA regardless of the APW+lo or PPW method, while our PBE-GGA [24] remains superior for the prediction of its bulk modulus. This can also be the case for the lattice parameter of the Si compound. From the result presented in Table 1 one can cautiously state that the WC-GGA [22] functional improves the lattice parameters over both the LDA [28] and PBE-GGA [24]. This statement is in agreement with the prediction of Wu and Cohen for the most of their studied compounds [22]. Our results show that lattice parameters calculated from WC-GGA [22] approximation are very close to experiment [33,34]. The calculated lattice parameter of the bulk Si, i.e., 10.28 a.u., will be used in the forthcoming Section 4 for simulating this compound as our substrate to mimic the effects of the lattice mismatch on the valence bands at the interface of the Pb/Si(111). Finally, keeping in mind that the value obtained for the pressure derivative of the bulk modulus depends on the range of fitting and as a result quantitative analysis may not be completely reliable, our result shows that the PBE-GGA [24] yields better pressure derivatives of the bulk moduli for the Pb and Si compounds than LDA [28] and WC-GGA [22].

4. Slab determination and optimization

The silicon substrate has been simulated using the calculated WC-GGA [22] lattice parameter in the previous Section 3 along the (111) crystallographic direction of the $Fd\bar{3}m$ space group. We have found that the Si(111) substrate, as shown in Fig. 1a, can be properly modeled by stacking eight Si bilayers, i.e., 16 Si monolayers. Surface formation energy, E_{Si}^f , per unit area and work function, ϕ_{Si} , were calculated for the various number of Si(111) bilayers, N_{Si} , and the results are listed in Table 2. Our result shows

Table 1

Lattice parameter, bulk modulus and pressure derivative of bulk modulus of Si (Pb) together with the method of calculations and exchange–correlation (XC) functional.

Method	XC-potential	a (bohr)	B (GPa)	B'
APW+lo	GGA-WC	10.28 (9.31)	93 (47.3)	4.23 (2.9)
APW+lo	GGA-PBE	10.34 (9.49)	88 (42.2)	4.16 (5.0)
APW+lo	LDA	10.22 (9.21)	96 (53.1)	4.37 (3.7)
PPW	LDA ^{a,b}	10.17 (9.16)	87 (52.6)	3.82 (5.2)
PPW	GGA ^{c,b}	10.32 (9.56)	87 (39.3)	— (4.7)
Exp. ^{d,e}		10.26 (9.36)	98 (43.2)	4.02(4.9)

^a Ref. [30].

^b Ref. [31].

^c Ref. [32].

^d Ref. [33].

^e Ref. [34].

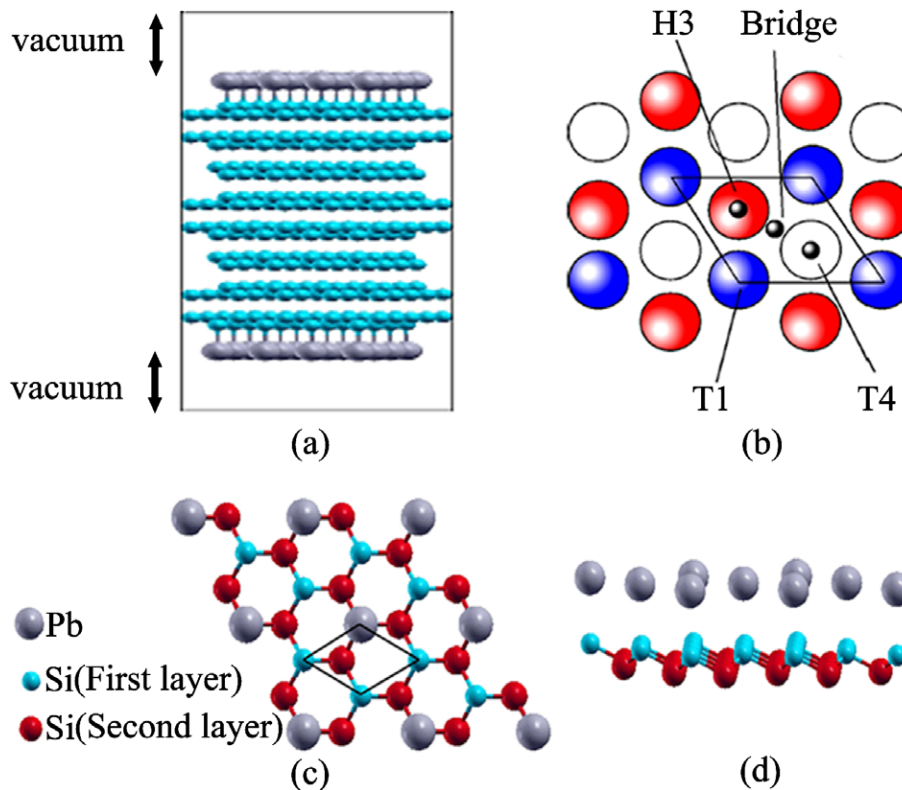


Fig. 1. (a) A Pb/Si(111) slab symmetrically immersed in a vacuum to illustrate atomic arrangement of Pb adatoms (larger grey spheres) on eight bilayers of Si(111) atoms (smaller light spheres). (b) A schematic view of Pb overlayers to illustrate T1 (top), H3 (hcp), T4 (fcc) and bridge (intermediate point of the T1 and T4 or T1 and H3) sites of the Pb adatoms on the Si(111) substrate. Top (c) and cross (d) views of the the most stable phase, T1, including a bilayer of Si(111) and a Pb overlayer located along the first layer of the Si(111) bilayer. The (1×1) -unit cell is outlined in (b) and (c).

Table 2

The work function, ϕ_{Si} (eV), and surface formation energy, E_{Si}^f (eV/bohr²), per unit area for the number of Si(111) bilayers (N_{Si}).

N_{Si}	ϕ_{Si} (eV)	E_{Si}^f (eV/bohr ²)
4	4.8484	1.3037
6	4.7179	1.3281
8	4.7295	1.3274
10	4.7299	1.3270

that the changes of the work function and the surface formation energy are less than 0.001 eV by adding more bilayers. The reliability of the obtained eight Si bilayers has been further investigated by comparing the bulk and surface electronic structures of the silicon. The total densities of states (DOSs) were calculated for both the bulk and surface of the Si. One expects to observe similar DOSs for the bulk of Si and the deepest atom from the surface of the Si(111). Our calculated surface DOS for $N_{\text{Si}} < 8$ predicts a wrong metallic behavior for the deepest Si atom. However, our result confirms that the DOS of the deepest Si layer approaches to the bulk DOS of the Si semiconductor by approaching N_{Si} to the eight bilayers. We avoid to present the DOSs of the substrate without the deposited Pb layers in this paper, as we just used them to ensure the validity of the Pb/Si(111) thin films calculations. The Pb layers were then deposited over and below the prepared Si(111) substrate to create a symmetric Pb/Si(111) slab. In order to simulate the Pb/Si(111) thin films, the created Pb/Si(111) slab, as shown in Fig. 1a, is symmetrically immersed in a vacuum as well. The symmetric thin films causes to add inversion symmetry which gives rise to speed up the interface relaxation. The vacuum thickness is defined to be the distance between the top edge of the slab and the bottom edge of its next neighbor. In order to determine the

vacuum thickness, total energy, work function and exerted forces on the surface atoms were calculated for the eight bilayers Si(111) structure versus various vacuum thicknesses. Our results, which are not presented here, show that 12 Å is sufficient for the vacuum thickness to avoid interactions with the nearest neighbors of the slab along the Cartesian z axis. The total energy, work function and forces were well converged in the vicinity of the obtained vacuum thickness – they did not show significant changes by increasing the vacuum thickness more than the value obtained. All the Si underneath atoms and Pb adatoms are allowed to fully relax by adjusting their heights, but not their coordinates parallel to the Si(111) surface. The top (T1), fcc (T4), hcp (H3), and bridge (B2 – intermediate point of the T1 and T4 or T1 and H3) sites, as corresponding phases for the deposition of the Pb layers on the Si(111) surface, are shown in Fig. 1b. In order to better visualize the most stable phase, top and cross views of the $(\sqrt{3} \times \sqrt{3})$ -unit cell of the T1 phase are illustrated in Fig. 1c and d, respectively. In this phase, as shown in Fig. 1d, the Pb adatoms were positioned along the first layer of the the first Si(111) bilayers. The (1×1) -unit cell as the interface structure for our full-potential all electron ab initio calculations is outlined in Fig. 1b and c.

5. Interface properties

5.1. Stability

In this section, we would make sure about the most preferable configuration. Total energies were then calculated for all of the illustrated structures in the Fig. 1b, apart from the bridge-site structure. The calculations were performed in the presence and absence of the spin–orbit coupling (SOC). Total energies relative to

the T1 site are presented in Table 3. Our result with and without SOC shows that the T1 site is the most stable structure among the other configurations. This result is in agreement with experiment [6]. The energy, as shown in Table 3, increases from T1 site to T4 site and decreases from T4 site to H3 site. The later trend, as shown in Table 3, is not affected by the spin–orbit interactions. The result shows that the effect of spin–orbit interactions, keeping the trend, is to shift downward total energies of all the sites. Our results are also quantitatively comparable with the pseudopotential calculations [35]. The energy of the bridge-site phase has not been calculated, since, as mentioned in the previous Section 4, B2 is an intermediate site of T1 and T4 sites or T1 and H3 sites. Therefore, one can most likely predict that the surface formation energy of the B2 site because of its symmetry is somewhere between T1 and T4 sites or between T1 and H3 sites. This prediction is in complete accord with the pseudopotential results [35]. We will concentrate only on the lowest-energy T1 site from later on in the remaining subsequent sections.

5.2. Si–Pb bond length

Bond length of Pb–Si, $d(\text{Pb–Si})$, at the interface of our slab has been calculated for the most energetically stable phase, i.e., T1 site, as shown in Table 3. The WC-GGA [22], PBE-GGA [24] and LDA [28] functionals were used for the exchange–correlation (XC) potential. Our full-potential results together with the pseudopotential results of the others [4] as well as experimental data [36] are listed in Table 4. The experimental data were obtained by surface X-ray diffraction based on the $\text{Pb/Si}(111)\text{--}(\frac{3}{-1} \frac{2}{1})$ model [36]. This model contains seven Pb adatoms in the $(\frac{3}{-1} \frac{2}{1})$ -unit cell [36]. Here we are only interested in the four of these lead atoms, since they are located in the top (T1) site as our predicted most stable phase [36]. The Pb–Si bond lengths for these four lead adatoms were measured [36] to be between $2.66 \pm 0.03 \text{ \AA}$ and $2.98 \pm 0.03 \text{ \AA}$ [36]. The covalent radii of silicon and lead are $R_{\text{co}}(\text{Si}) = 1.18 \text{ \AA}$ and $R_{\text{co}}(\text{Pb}) = 1.47 \text{ \AA}$, respectively, and the bulk radius of lead is $R_{\text{bu}}(\text{Pb}) = 1.75 \text{ \AA}$ [36]. Thereby the covalent bond lengths of Pb–Si have to be between $R_{\text{co}}(\text{Si}) + R_{\text{co}}(\text{Pb}) = 2.65 \text{ \AA}$ and $R_{\text{co}}(\text{Si}) + R_{\text{bu}}(\text{Pb}) = 2.93 \text{ \AA}$, viz. $2.65 \text{ \AA} < d_{\text{co}}(\text{Pb–Si}) < 2.93 \text{ \AA}$ [36]. Our

LDA full-potential result, as shown in Table 4, is in excellent agreement with the LDA pseudopotential result of the others [4]. The LDA results, as shown in Table 4, are very close to the strongest covalent bond length, i.e., $2.65 \text{ \AA}$. The result demonstrates that the PBE-GGA and WC-GGA give larger bond lengths which result in weaker Pb–Si bonds than LDA. This confirms a trend shown in Table 1 in which the LDA underestimates but GGA overestimates lattice parameters compared to experiment. The weaker PBE- and WC-GGA bonds, as shown in Table 4, are still in the covalent bond length interval, i.e., $[2.65, 2.93 \text{ \AA}]$. The predicted covalent bonds by LDA and GGA show that the competition between the Pb–Si and Pb–Pb interactions gives rise to saturate all the dangling bonds of the silicon substrate. Finally, it worths to mention that the trend seen in Table 1 is reversed here, as shown in Table 4, between PBE-GGA and WC-GGA in which PBE-GGA predicts lattice parameter larger than WC-GGA. In contrast to the lattice parameter trend, here, as shown in Table 4, the Pb–Si bond length within the WC-GGA is slightly larger than the Pb–Si bond length within the PBE-GGA. Normal and in-plane Pb–Si distances, viz. d_z and d_{xy} , were calculated employing pseudopotential method for the $\sqrt{7} \times \sqrt{3}$ mixed phase obtained from coadsorption of Pb and Sn on the $\text{Si}(111)$ surface. The Pb–Si bond lengths, $d = (d_{xy}^2 + d_z^2)^{\frac{1}{2}}$, were found to vary from $2.77 \text{ \AA}$ to $3.33 \text{ \AA}$ depending on the positions of silicon atoms with respect to the lead adatoms for the (Pb, Sn)/ $\text{Si}(111)$ $\sqrt{7} \times \sqrt{3}$ phase [19]. The Pb–Si bond length of the later $\sqrt{7} \times \sqrt{3}$ model with less in-plane Pb–Si distance can be more comparable with the bond length obtained from our top site phase. The less in-plane distance and its corresponding normal distance were calculated to be $d_{xy} = 1.79 \text{ \AA}$ and $d_z = 2.11 \text{ \AA}$, respectively, which resulted in Pb–Si bond length $d = (1.79^2 + 2.11^2)^{\frac{1}{2}} = 2.77 \text{ \AA}$ [19]. The later bond length, $2.77 \text{ \AA}$, which is in the above mentioned covalent bond length interval, i.e., $[2.65, 2.93 \text{ \AA}]$ and closer to our WC-GGA result, asserts that the Pb–Si forms a very strong bond [19].

5.3. Work function and surface formation energy

We calculated the work function, ϕ (eV), as the minimum energy required to liberate an electron from the Fermi level (E_F) to a point with negligible kinetic energy at the center of the vacuum of the slab. The calculations were performed using the following formula:

$$\phi = E_{\text{vac}} - E_F, \quad (1)$$

where E_{vac} is estimated by the averaged electrostatic Coulomb potential at the midpoint of the vacuum of the slab and E_F is the corresponding Fermi energy. The work function is a sensitive parameter to the surface conditions, since the liberated electron must move through the surface. The work functions were calculated at the interface of the slab with and without SOC and the results are presented in Table 5. The result shows that the SOC causes to reduce the work function. The pseudopotential results for the clean $\text{Pb}(111)$ surface [37,38] are also given in Table 5 within two different exchange–correlation functionals. The comparison shows that the various versions of GGA can result in slightly different values. Since the pseudopotential work functions are available for the hypothetical freestanding $\text{Pb}(111)$, we also added our results without $\text{Si}(111)$ substrate in Table 5 for the comparison. Our freestanding work function is calculated for the second layer of the $\text{Pb}(111)$. A single lead layer without any substrate may not constitute a meaningful stable physical system. Our $\text{Pb}(111)$ work function which is in good agreement with the pseudopotential result would be compared with the results of our actual $\text{Pb/Si}(111)$ slab to elucidate the role of $\text{Si}(111)$ substrate. An overlook in Table 5 shows that our calculated work functions taking silicon substrate into account are in the same order of magnitudes when compared to the free-

Table 3

Relative total energies with respect to the energy of the T1 phase, $E(\text{T1})$, in the presence and absence of the spin–orbit coupling (SOC). Here the $E(\text{T1})$ is arbitrarily chosen to be zero.

SOC	T1 (eV)	T4 (eV)	H3 (eV)
No	0.00	0.41	0.32
Yes	0.00	0.25	0.20
No ^a	0.00	0.35	0.18

^a Ref. [35].

Table 4

Pb–Si bond length, $d(\text{Pb–Si})$, together with the method of calculations and exchange–correlation (XC) functional.

Method	XC-potential	$d(\text{Pb–Si})$ (Å)
FP-APW+lo	WC-GGA	2.72
FP-APW+lo	PBE-GGA	2.70
FP-APW+lo	LDA	2.68
PPW ^a	LDA	2.66
EXP. ^b		$2.66 \pm 0.03 \leq d \leq 2.98 \pm 0.03$

^a Ref. [4].

^b Ref. [36].

Table 5

The work function, ϕ (eV), and surface formation energy, E^f (eV/Å²), per (1 × 1)-unit cell for the Pb at the interface (1 ML coverage) of the Pb/Si(111) together with the method of calculations, kind of exchange-correlation (XC) functionals and interactions as well as existence or extinction of the Si(111) substrate. Since most of the presented pseudopotential results of the others in this table are for the hypothetical freestanding Pb(111), our FP-APW+lo results without Si(111) substrate are also included for the comparison.

Method	XC-potential	SOC	Si(111)	ϕ (eV)	E^f (eV/Å ²)
FP-APW+lo	WC-GGA	No	Yes	4.72	−0.59
FP-APW+lo	WC-GGA	Yes	Yes	4.65	−0.61
FP-APW+lo ^a	WC-GGA	No	No	4.23	0.02
PPW ^b	LDA	No	Yes	–	−0.47
PPW ^c	PW-GGA	No	No	4.07	–
PAW-PPW ^d	GGA92	No	No	3.83	0.03

^a These freestanding Pb(111) values are given for $N=2$. For $N=1$, the hypothetical freestanding film is so unstable that may no longer be considered as a physical system even within a pure theoretical study.

^b Ref. [4].

^c Ref. [37].

^d Ref. [38]. The given surface formation energy in the Ref. [38], 0.372 eV, is converted from eV to eV/Å². Hence the value of 0.372 eV is divided by the area of the (1 × 1)-unit cell, 12.804 Å².

standing full- and/or pseudopotential results. From the later point, it appears that the Si substrate does not significantly influence the work functions. The later point confirms the observation of Kirchmann et al. [15], where they [15] found the agreement between their experimental QSE observation for the actual Pb/Si(111) and the pseudopotential results [14] for the hypothetical freestanding Pb(111).

The surface formation energy per unit area at zero temperature is defined as [39]:

$$E^f = \frac{1}{2A} (E_{\text{slab}} - N_{\text{Si}} E_{\text{Si}}^{\text{bulk}} - N_{\text{Pb}} E_{\text{Pb}}^{\text{bulk}}), \quad (2)$$

where E_{slab} is the total energy of the slab, N_{Si} (N_{Pb}) and $E_{\text{Si}}^{\text{bulk}}$ ($E_{\text{Pb}}^{\text{bulk}}$) are the number and bulk energy of the Si (Pb) atoms in the unit cell, respectively. Here A is the area of the (1 × 1)-unit cell. The factor $\frac{1}{2}$ is used in the above formula, since the slab has two surfaces, as shown in Fig. 1a, due to the embodied inversion symmetry. In principle one can use the above equation as the standard method to calculate the surface formation energy. However, it is well-known that in practice the surface formation energy employing Eq. (2) diverges [40,41]. In order to overcome the divergence problem of the standard Eq. (2), in this paper we have used the following formula [40] to calculate the surface formation energy:

$$E^f = \frac{1}{2A} (E_{\text{slab}} - N_{\text{Si}} \Delta E_{\text{Si}}^N - N_{\text{Pb}} \Delta E_{\text{Pb}}^N), \quad (3)$$

where $\Delta E_{\text{Si}}^N = E_{\text{Si}}^N - E_{\text{Si}}^{N-1}$ and $\Delta E_{\text{Pb}}^N = E_{\text{Pb}}^N - E_{\text{Pb}}^{N-1}$. We have already used Eq. (3), with $N_{\text{Pb}} = 0$, in the previous Section 4 to calculate the surface formation energy of the pure Si(111) substrate. Our calculated surface formation energies in the presence and absence of the spin-orbit coupling (SOC) along with the pseudopotential results for both of the actual Pb/Si(111) interface [4] and hypothetical clean Pb(111) surface [38] are given in Table 5. Our result shows that the effect of SOC is small as well. Spin-orbit (SO) coupling breaks down the space inversion symmetry by imposing a preferable direction, which gives rise to spin splitting. The inversion symmetry can be also broken at an interface or a surface. The Rashba-effect [42] on metallic surfaces is a phenomenon, which originates both from spin-orbit coupling and the lack of inversion symmetry at surfaces [43,44]. A Rashba-type spin-orbit splitting with no coverage dependency was found for quantum well states formed in ultrathin Pb films on Si(111) [18]. Hugo Dil et al. [18] found for the states at 0.15 and 0.4 eV this splitting to be 14 and 15 meV,

respectively. We found the same splitting in the vicinity of Fermi level for the valence bands. The SO coupling within this small splitting, as shown in Table 5, causes to reduce the work function and surface formation energy by 0.07 eV and 0.02 eV/Å², respectively. As shown in Table 5, the surface formation energy is a negative value taking substrate into account, i.e., for the actual Pb/Si(111) case. On the contrary it is a tiny positive value without substrate, i.e., for the hypothetical freestanding Pb(111) case. The obtained negative value when compared with the tiny positive value confirms that the Pb/Si(111) is energetically more favorable to form that particular structure than the clean Pb(111) surface. The later point is in complete accord with the pseudopotential results, as can be seen from Table 5. Therefore, the effect of substrate is obviously to make the Pb layers energetically stable.

5.4. Density of states (DOS)

Total and partial densities of states (DOSs) were calculated at the interface for both of the underneath Si and the overlayer Pb atoms. The calculations were performed in the absence and presence of the spin-orbit coupling (SOC) for the slab shown in Fig. 1a. Total DOSs with and without SOC, which are not shown here, were compared. Despite the large atomic number of Pb, the comparison shows that the effect of SOC is not very significant on the valence electronic structure and can be ignored. This is in accordance with the experiment in which the magnitude of the Rashba-type spin-orbit splitting were found for this thin films to be so small that can be detected by the spin integrated angle resolved photoemission (ARPES) method [18]. The spin splitting originates from all of the layers and thereby should be calculated by integrating over the entire layers [18]. The spin splitting is small in Pb/Si(111) ultrathin films, because the signs of the splittings of the Pb–vacuum and Si(111)–Pb interfaces are opposite in direction due to the larger phase shift at the Pb to the vacuum interface than the phase shift at the Pb–Si(111) interface [18]. Thus the total splitting which partly cancels each other results in an ignorable spin-orbit coupling effect in Pb/Si(111). Therefore from now on we do not include the spin-orbit interactions in all of our surface calculations. Partial s and p DOSs are shown in Fig. 2a and b for Pb and in Fig. 2e and f for Si atoms at the interface of the slab, respectively. The result shows that the s states of the Pb and Si atoms are almost distributed over the energy interval of (−10, −5) eV, while their p states are nearly in the (−4, $E_F = 0$) eV energy window. This demonstrates that the s states are lower in energy than the p states for both of these Pb and Si atoms. Consequently, the s states of Pb and Si are more localized than their p states, which are closer to the Fermi level when compared with the former s states. Since the p orbital manifest itself more closely to the Fermi level, the p states show more itinerant character than the s states. Hence, the p electrons play more important role than the s electrons in the Pb–Si bonding. Now we can discuss the kinds and qualitatively the strengths of the Pb–Si bonds, keeping three points in mind: (i) s (p) electrons are more localized (itinerant) than p (s) electrons for both of the Pb and the Si atoms and (ii) below the Fermi level interval energy (−10, −5) eV of s states do not overlap interval energy (−4, 0) eV of p states for both of the Pb and Si atoms, as well as (iii) Pb sites on-top of Si site vertically along z direction, as shown in the top site configuration in Fig. 1b. The result shows that the 6s states of Pb adatoms are weakly hybridized with the 3s states of the Si substrate. According to the above point (ii) the energy window of 3s Si well overlaps the energy window of 6s Pb, as shown in Fig. 2a and e. One may then promptly speculate that the 6s Pb states can be well hybridized with the 3s Si states. However, according to the point (i) both of the 6s and 3s states are nearly localized and so do not substantially contribute in bonding. Thereby the overlap of the 3s Si and 6p Pb DOSs can only result

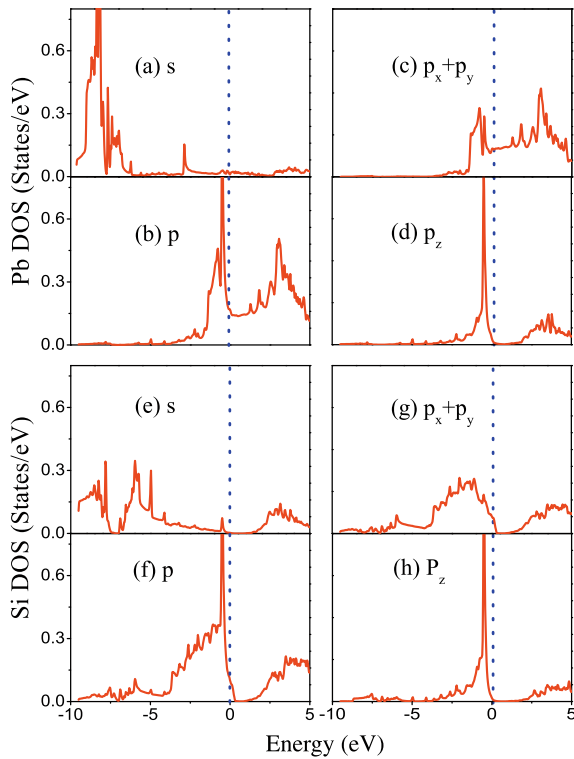


Fig. 2. Partial densities of states for (a) s Pb, (b) p Pb, (c) p_{xy} Pb, (d) p_z Pb, (e) s Si, (f) p Si, (g) p_{xy} Si, (h) p_z Si.

in an extremely weak σ -like bond, which may be due to its weakness neglected. The 6s Pb states are even much more weakly bonded to the p Si states compared to the former extremely weak σ -like Pb–Si bond. According to the property (ii) energy windows of 6s Pb and 3s Si do not overlap, because the s-Pb states are over $(-10, -5 \text{ eV})$, while the p-Si states are in $(-4, 0 \text{ eV})$, as can be clearly seen from Fig. 2a and f. This in turn gives rise to even weaker bond between 6s Pb and p Si states than the last 6s Pb and 3s Si bond. This is also the case, as shown in Fig. 2b and e, for the p Pb and s Si DOSs. Till now, there are no any evidence of the appearance of Pb–Si bonding. Let us then discuss the bonding states by concentrating solely on the most important p states of the Pb and Si. First, the result shows that the energy window of p Pb well overlaps the energy window of p Si, as shown in Fig. 2b and f. Second, according to the above mentioned (i) point, the p Pb and Si electrons are more itinerant than their s electrons. Third, there are a lot of p Pb and p Si states at an energy very close to the Fermi level, see two sharp peaks in Fig. 2b and f. Therefore, the p states of the Pb and Si atoms can be well hybridized with each other. In order to determine the kind of Pb–Si bond, the p states of Pb and Si were further decomposed, as shown in the right panel of the Fig. 2, to their partial p_z and p_{xy} states. The distributions of the p_z and p_{xy} DOSs of both Pb and Si atoms show that they are completely hybridized with each other. Here, the z axis is taken to be perpendicular on the surface and x, y are parallel to the surface. The Pb overlayer atom is just located above the Si underneath atom along the z-Cartesian coordinate in the top site configuration, as shown in Fig. 1b. Hence, one lobe of the p_z orbital of Si underneath atom overlaps with one lobe of the p_z orbital of Pb overlayer atom. Consequently, hybridization of p_z orbitals of Pb and Si results in a σ -bond. The strength of the p_z Pb and p_z Si σ -bond can be estimated by observing that the two sharp peaks at an almost single energy nearby the Fermi level originate from the p_z Pb and p_z Si DOSs, as shown in Fig. 2d and h. This can be taken as an indication to the fact that there is a tightly σ -bond between the p_z Pb and p_z Si at the interface. This result

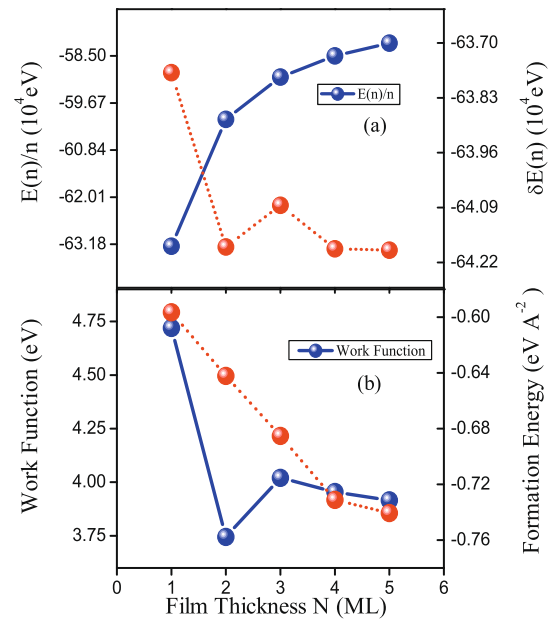


Fig. 3. (a) Total energy per layer (solid line) and energy differences (dotted line) of Pb/Si(111) versus number of monolayer (ML) coverages, N . (b) Surface formation energy (dotted line) per (1×1) -unit cell and work function (solid line) of Pb/Si(111) versus number of monolayer (ML) coverages, N .

is in excellent agreement with the pseudopotential result of the others [19] for Pb/Si(111) $\sqrt{3} \times \sqrt{7}$ phase, in which a strong hybridization between the p_z Pb orbital and the p_z dangling bonds of the substrate had been reported. As shown in Fig. 2c and g, two lobes of the p_{xy} Pb overlap with two lobes of the p_{xy} Si as well, which gives rise to a π -bond. The strength of the π -bond cannot be so large, because the p_{xy} Pb and p_{xy} Si DOSs are broadened, as shown in Fig. 2d and h, in contrast to the sharp narrow p_z Pb and Si DOSs. Thereby, a feeble π -bond is constituted by the p_{xy} Pb and p_{xy} Si at the interface. We conclude this discussion by stating the following points. There is not a pure π or σ state in the Pb–Si bond. Instead there is a mixed state composed of π and σ states in the Pb–Si bond. The combination of these two feeble π and strong σ bonds can result in a strong covalent Pb–Si bond. The later point is consistent with our last discussion in Section 5.2 regarding the bond length of Pb–Si. In summary, the Pb overlayer can be strongly absorbed by the Si substrate.

6. Thin films properties

6.1. Total energy and energy differences

Total energy per number of layers, $E(N)/N$, is shown in Fig. 3 a as a function of the film thickness, N . The result shows that the $E(N)/N$ increases by N and asymptotically approaches to a constant value. The asymptotic behavior can be realized, if we notice that the motion of Pb electrons is confined in one direction while it remains free to move on the surface. The later point itself is due to the fact that the Pb/Si(111) is a prototype of the metal-on-semiconductor system with a high Schottky barrier against penetrating valence electrons of Pb into the substrate [45,46]. The confinement in the nanoscale causes wave vector k to quantize in a perpendicular direction on the surface, which results in discrete energy levels associated with the so-called quantum-well (QW) states. The simplest way to describe this confinement is then by the one dimensional QW [47]. In a direction normal on the surface there is no translation symmetry. The potential in the xy plane is periodic

and potential in the z direction represents the QW. The energy levels in the free electron approximation are given by:

$$E_n = \frac{\hbar^2}{2m} (k_{\parallel}^2 + k_{\perp}^2) = \frac{\hbar^2}{2m} (k_x^2 + k_y^2) + \frac{\hbar^2 \pi^2 n_z^2}{2md^2}, \quad (4)$$

where m is the free electron mass and d is the thickness of the film. The above equation clearly shows that the total energy approaches to a constant value by increasing the film thickness. This equation elucidates recognition of the asymptotic behavior shown in Fig. 3a. Monolayer per atom energies $\frac{E(N)}{N}$ were calculated for the freestanding Pb(111) slab as a function of monolayers N [13]. In agreement with our above discussed asymptotic behavior for the actual case of Pb/Si(111), it was shown that $\frac{E(N)}{N}$ of the hypothetical freestanding Pb(111) system gradually approaches a constant value [13].

The energy difference between two successive layers is defined as:

$$\delta E(N) = E(N) - E(N-1). \quad (5)$$

The $\delta E(N)$ versus number of layers, N , is also shown in Fig. 3a. The result shows that the energy difference oscillates as a function of the number of monolayer (ML) coverages with the period of $\lambda = 2$ ML. The oscillation is attributed to the quantum size effect (QSE). Similar QSEs were detected by the pseudopotential calculations for the freestanding Pb(111) thin films [13]. One expects that the underlying physics behind of the observed QSE in the $\delta E(N)$ quantity, as shown in Fig. 3a, can be related to the electronic structure of the system. We would postpone this point to a more appropriate time in Section 6.3, where the electric field gradient (EFG) as an extremely sensitive quantity to the valence electron charge density distribution is discussed.

6.2. Work function and surface formation energy

The work function, as discussed in the previous Section 5.3, has been calculated utilizing Eq. (1). The result is presented in Fig. 3b as a function of Pb thin films thickness for the actual Pb/Si(111) slab. The result shows an oscillatory behavior of the work function versus the film thickness with the period of $\lambda = 2$ ML. The oscillation is quickly damped and the work function is going to be converged through a few number of Pb layers. The work function oscillation was previously attributed to the quantum size effects (QSE) and predicted [14] by using a pseudopotential calculation for the case of a freestanding Pb slab. The pseudopotential result [14] for the hypothetical clean Pb(111) thin films shows that the oscillation is much more slowly damped and the work function is converged at a very large number of Pb layers when compared with our full-potential results for the real Pb/Si(111) case. The different oscillatory behavior might be related to the effect of the Si substrate. But one has to notice that the discrepancy may come from the two different approaches. Thus we calculated the work function for the clean Pb(111) surface within our full-potential method as well. The result as shown in Fig. 4 shows that our full-potential work functions are also no longer rapidly damped for the clean Pb(111) case in agreement with the pseudopotential result. This verifies that the oscillatory behavior discrepancy indeed results mainly from the existence and extinction of the Si substrate and not from the method of calculations. This authenticates that the occurred QSE in the work function is influenced by that of Si(111). The effect of Si substrate on the work function confirms the calculation of Dil et al. [16], where they [16] similarly found that unlike for Pb on graphite, the Pb overlayer lattice structure is influenced by the Si(111). The effect of our Si(111) substrate on both of the geometry and electronic structures of the Pb overlayer atoms is also in agreement with the influence of the

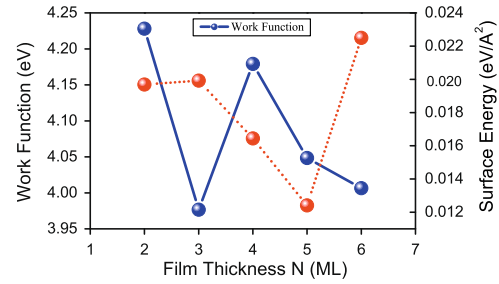


Fig. 4. Surface formation energy (dotted line) per (1×1) -unit cell and work function (solid line) of freestanding Pb(111) versus number of monolayer (ML) coverages, N .

Cu(111) substrate on the geometry structure of the Pb(111) layers reported by Materzanini and coworkers [13]. In the later work, the effect of substrate was studied in a tricky manner without including directly Cu(111) underneath atoms by compressing the supercell dimensions in the surface plane by 3.3% [13]. They found that the explicit inclusion of the Cu(111) substrate would be highly desirable to explain the remaining quantitative differences between theory and experiment [13]. It seems that the later result contradicts our last finding in Section 5.3. We found in Section 5.3 that the order of magnitude of the work function could not be significantly affected by the substrate in agreement with the observation presented in Ref. [15]. The contradiction can be resolved if we notice that the work function is not so sensitive that its order of magnitude, but its oscillatory behavior, can be affected by the substrate. The electric field gradient (EFG) is much more sensitive quantity than the work function to the surface states. We shall show in subsequent Section 6.3 that the EFG is so sensitive to the substrate that not only its oscillatory behavior but also its value can be affected by the Si substrate.

We calculated the surface formation energy versus the film thickness, N , based on the Eq. (2). The result, which is not shown here, varies linearly with respect to N . The linear variation is consistent with the first scheme described in Ref. [40]. We then recalculated the surface formation energy versus N according to the Eq. (3). The result, as shown in Fig. 3b, manifests a deviation from the linear behavior on going from fourth to the fifth layer. The deviation is consistent with the third scheme discussed in Ref. [40]. Here we did not use the fourth scheme given in Ref. [40]. According to the fourth scheme the bulk energies in the standard Eq. (2) can be found by taking the slope of a fitted straight line to all of the slab total energies versus N . It is generally believed [40,41,48] that the divergence problem disappears employing the fourth scheme for smaller N . However, in this paper we aim to elucidate the role of substrate. The goal of this paper can be achieved even by the third scheme which is identical to the the Eq. (3). Therefore we calculated the surface formation energy versus N for the clean Pb(111) surface. The result is shown in Fig. 4 which can be compared with the surface formation energy of the actual Pb/Si(111) slab as shown in Fig. 3b. Furthermore the clean Pb(111) surface formation energy behaves completely different from that of its supported slab by the Si substrate. The different behavior demonstrates the effect of Si on the surface formation energy.

6.3. Electric field gradient (EFG)

The electric field gradients (EFGs) [49,50] were calculated at the Pb sites for our slab. The main component of the EFG tensor, V_{zz} , is presented as a function of film thickness, N , in Fig. 5. The sign of the V_{zz} is negative, as shown in Fig. 5, for all the layers. The result shows that for 1 monolayer coverage, i.e., $N = 1$, the absolute value of the V_{zz} is a very large number. As shown in Fig. 5, the $|V_{zz}|$ is dras-

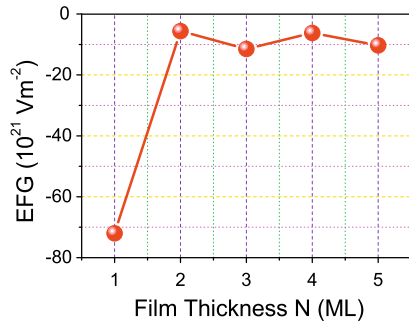


Fig. 5. The main component of the electric field gradient, V_{zz} , versus number of monolayer (ML) coverages, N .

tically reduced on going from first to the second layer. Despite the reduction is significant, the $|V_{zz}|$ values are still considerably large for $N \geq 2$. Our result shows that for the larger N the V_{zz} oscillates with respect to the number of monolayer (ML) coverages. The period of oscillation here is also $\lambda = 2$ ML. Therefore our ab initio calculations clearly elucidate the quantum size effect (QSE) in the electric field gradient (EFG). In order to interpret the above observation shown in Fig. 5a cartoon like electron charge distribution is illustrated in Fig. 6. The electric field gradient (EFG) is an extremely sensitive quantity to the deviation from the spherical valence electron charge density distribution [50–53]. The EFG, as shown in Fig. 6b, is zero for a spherical charge density distribution, which is the case for the cubic or higher point group symmetry. The EFG, as shown in Fig. 6a and c, can be nonzero at an atomic site depending on its point group symmetry for an aspherical charge density distribution, which is the case for our Pb sites. The EFG can be calculated within the APW method by $V_{zz} \propto \int [\rho(r)Y_{20}]/r^3 dr$ [54]. The EFG can be nonzero for an aspherical charge density distribution and obtained directly from the $l = 2, m = 0$ component of the potential expansion inside the Muffin-tin spheres [53]. The sign of the EFG depends on the orientation of the anisotropic charge density distribution. The sign is defined to be negative (positive) for the charge distribution shown in Fig. 6a and c. Our surface Pb atoms for 1 monolayer (1 ML) coverage are only supported by their underneath Si(111) substrate atoms, since for $N = 1$ there is no more Pb overlayer on the first deposited Pb layer. The Pb–Pb bonds in the (x, y) -plane compensate each other. However, there is no Pb–Pb bonds along the z Cartesian direction to compete with the Pb–Si bonds to compensate them. The Pb–Si bond, as discussed in Section 5.2, is a strong covalent bond. The uncompensated covalent Pb–Si bond results in a dramatic electron charge density deformation along the z direction.

Consequently one can, according to the Fig. 6a, first expect that the sign of the EFG is negative and second its value is dramatically large. These expectations are in excellent agreement with our result as shown in Fig. 5. For 2 ML coverages the second Pb layer is supported by the first Pb layer. The Pb–Pb bonds, which are again compensated in the (x, y) -plane, are much weaker [36] than the Pb–Si bonds along the z Cartesian direction. The weaker bonds results in a less charge density deformation which gives rise to a less value of the EFG at the Pb site for the second layer. The later point clearly explains the significant drop of the EFG on going from $N = 1$ –2 as shown in Fig. 5. Although the EFG is drastically reduced by adding the second layer, it can be still large due to the anisotropy of the Pb surface. The anisotropy comes from the fact that for $N = 2$ there is no Pb–Pb bond over the second layer to compensate the Pb–Pb bonds between the first and second layers.

Here we would back to the postponed point to clarify that the source of the quantum size effect (QSE) originates from the electronic structure of the system. In order to reach the purpose one can first make a connection between the EFG and the density of states (DOS). Fortunately an astonishing connection was previously made [55] between the EFG and the total DOS at the E_F , N_F . There it was found an approximately linear relation between the main component of the electric field gradient and the total density of states at the Fermi level, viz. $V_{zz} \propto N_F$, in the $CeIn_3$ compound [55]. Therefore from one side there can be a relation between the V_{zz} and the N_F in our thin films. On the other hand here we have shown that the V_{zz} oscillates as a function of film thickness. Consequently, there might be a relation between QSE and the electronic structure, or more specifically, between QSE and the total DOS at the Fermi level (N_F). Our finding regarding the later relation between N_F and the electron spillage length into the vacuum and as a result between N_F and the step height of the layer N [9]. This illuminates the underlying physics behind of the quantum size effects (QSEs). Our EFG study shows that the oscillations of a sensitive physical quantity with respect to the film thickness originates from the beating of the valence electron charge density distributions (VECDD). The VECDD deformation can change layer by layer depending on the bonds strength. The effect of the Si substrate, which is shown to be of significant importance in the strength of the bonds, plays an important role in the QSE. One suspects that the freestanding model can be applicable for those kinds of the thin films that the strength of their substrate–overlayer bonds are comparable with the strength of their overlayer–overlayer bonds. The later point is not the case for our thin films, since the strengths of the Pb–Si and Pb–Pb bonds are not comparable [36]. In conclu-

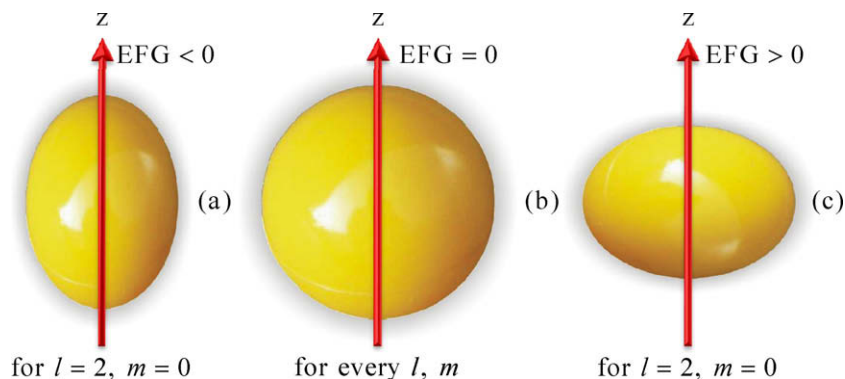


Fig. 6. A schematic representation of electron charge density distribution (ECDD). (a) A longitudinal ECDD deformation can yield a nonzero negative value for the main component of the electric field gradient, V_{zz} , for angular $l = 2$ and azimuthal $m = 0$ quantum numbers. (b) An undeformed spherical ECDD yields a zero value for the V_{zz} , for every l and m . (c) A transverse ECDD deformation can yield a nonzero positive value for the V_{zz} , with $l = 2$ and $m = 0$.

sion our QSE calculation shows that our system is not an ideal paradigm to freestanding films.

7. Conclusions

The effects of Si(111) substrate were investigated on the physical properties of the Pb/Si(111) thin films. The investigations were performed within the density functional theory (DFT) employing the augmented plane waves plus local orbital method (APW+lo). We used the PBE-GGA and WC-GGA for the exchange-correlation functional. Our result shows that the WC-GGA is more reliable to study this system. We included spin-orbit interactions in our the Kohn-Sham Hamiltonian. Our result shows that the effect of spin-orbit coupling is not of significant importance for this thin film. Several structures were considered for the slab. Our ab initio full-potential calculation shows that the top site (T1) is the most stable phase. The electronic structures at the interface of the most stable T1 phase were studied. The study shows that the Pb and Si at the interface are strongly bonded by σ and weakly by π bonds. The strong covalent Pb-Si bond is consistent with experiment. In order to elucidate the role of Si substrate the Quantum size effects (QSEs) were studied for the T1 configuration. The study is carried out by calculating the work function and surface formation energy as well as the electric field gradient (EFG) at the Pb sites. The calculations were performed as functions of the thin film thickness (N). We connected the oscillatory behavior of our calculated physical quantities to the electronic structures at various layers of the system. The connection has been made by the concept of the electric field gradient (EFG) as an extremely sensitive quantity to the electronic structure. The EFG provides a reliable approach to undertake the underlying physics behind of the QSE. Therefore the electric hyperfine interaction has merit as a measurement method to be further used in thin films. Our result shows that the effect of Si substrate depends on the sensitivity of the physical quantities to the valence electron charge densities. The Si substrate can be considerable for the sensitive quantities such as electric field gradients (EFG), while it can be ignored for less sensitive quantities such as the value of the work function.

Acknowledgments

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